

## CAMPANIAN POTTERY; THE NATURE OF THE BLACK COATING

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### 1. Introduction

Campanian pottery is generally defined as being a black-coated ware which was produced from the fourth to the first centuries B.C. in the western area of the Mediterranean basin (Morel 1980). During the second and the first centuries B.C. three productions can be clearly distinguished by their extensive distribution and by their particular typological characteristics. The type A has a red paste, the type B a creamy-yellow paste and the type C a grey paste (Lamboglia 1952). In all three cases we are dealing with ceramics produced in Italy which were very widely exported in the western and even the eastern Mediterranean. The type B includes several varieties which correspond to productions coming from different regions of Italy. This is not the case for the type A which originates in the Naples region and the type C which originates in Sicily. A variety of Campanian B, of very high quality, seems to have been produced in Etruria where it is well represented in the excavated material from Cosa (Taylor 1977). Another variety, very prevalent amongst the exports, would have been produced in the Calès region, a site already known for its relief productions of the second half of the third century and the beginning of the second century B.C. (Pagenstecher 1909).

The characteristics of the paste's colour of the Campanian types A and B are due to differences in the percentages of CaO (Picon 1977), the type A having a non calcareous or moderately calcareous paste (3–5 wt% CaO), the type B being very calcareous (8–11 wt% CaO) according to the nomenclature of Maniatis and Tite (1978, 1978/79).

At Calès itself, two types of paste are found. One is very calcareous (13.6 wt% CaO, as mean for 8 analysed specimens) and appears at a later period on the site and concerns above all the so-called pre-sigillata pottery. The other paste, less calcareous (9.0 wt% CaO, as mean for 25 analysed specimens), corresponds to the vast majority of black-coated wares from this site. More recent analyses carried out on twenty specimens (from the collection of M. Picon) of the most frequently found type of paste at Calès confirm these characteristics (table 1).

The present study deals with the characteristic black coating of these twenty objects; all

Table 1 Chemical analysis by X-ray fluorescence of Campanian sherds, all without gloss

wt. %	3 individual sherds from Min. Petr. Institut, Fribourg			20 sherds from Min. Petr. Institut, Fribourg			25 sherds from M. Picon		
	KP 1	KP 5	KP 10	$\bar{x}$	$s$	$s_{rel}(\%)$	$\bar{x}$	$s$	$s_{rel}(\%)$
SiO <sub>2</sub>	59.97	58.07	58.37	58.94	1.07	1.8	59.5	1.2	2.0
TiO <sub>2</sub>	0.75	0.77	0.81	0.77	0.02	2.6	0.79	0.05	6.3
Al <sub>2</sub> O <sub>3</sub>	16.78	17.42	17.90	17.08	0.42	2.4	17.4	0.56	3.2
Fe <sub>2</sub> O <sub>3</sub>	5.89	4.94	6.35	6.27	0.22	3.5	6.60	0.28	4.2
FeO	0.25	0.97	0.33	0.18	0.25	138.9	n.d.	n.d.	—
MnO	0.10	0.10	0.10	0.11	0.01	9.1	0.11	0.01	9.1
MgO	3.13	2.74	2.99	3.06	0.25	8.2	3.37	0.34	10.1
CaO	9.66	10.53	8.60	9.11	1.34	14.5	9.0	1.36	15.1
Na <sub>2</sub> O	0.99	0.98	1.01	0.95	0.08	8.4	n.d.	n.d.	—
K <sub>2</sub> O	2.95	2.98	3.25	3.03	0.09	2.9	2.96	0.14	4.7
P <sub>2</sub> O <sub>5</sub>	0.28	0.69	0.67	0.52	0.24	46.1			
	100.75	100.19	100.38						
loss	2.06	1.98	2.13						
ppm									
Ba	395	405	453	418	62	14.8			
Zr	177	167	170	169	9	5.3			
Sr	315	319	312	300	45	15.0			
Rb	141	140	143	143	8	5.6			
Zn	95	93	108	99	9	9.1			
Cu	33	41	45	42	14	33.3			
Ni	64	59	65	64	5	7.8			
Cr	103	99	102	104	6	5.8			

$s$  = standard deviation,  $s_{rel}$  = relative standard deviation,  $\bar{x}$  = mean value, n.d. = not determined.

sherds were analysed by X-ray diffraction technique, X-ray fluorescence, five studied by scanning electron microscopy (nos. 1, 4, 6, 11, 17), three by microprobe (1, 5, 10) and eleven by direct X-ray diffraction of the surface (1–7, 10, 11, 14, 17, 19).

Similar glossy clay coatings have been widely used in antiquity for pottery production (e.g. Attic black vases, red Terra Sigillata). However, only a few authors presented *complete chemical analyses* of these coatings. Our results are therefore compared with the conclusions of Pavicevic (1974), Tite *et al.* (1981) on the black coatings and of Willgallis and Heyer (1976), Tite *et al.* (1981) on the red coatings.

## 2. Experimental

**X-ray fluorescence** The sherds were analysed without the coating. *Glassy pill*: 900 mg of the heated powder (1 hour at 1100°C) were mixed with 4.5 g of a Li-tetraborate-, Li-carbonate-, La-Oxide-mixture, and fused in a PtAu crucible during 10 min at 1050°C. *Powder pill*: 4.5 g of the dried powder (110°C for 24 h) were mixed with 8 drops of a 2.5% Mowiol-solution and pressed under vacuum at 2.5 tons pressure. For precision and accuracy of the method see Kuepfer and Maggetti (1978).

*Electron microprobe* Microprobe: SEMQ-ARL combined with EDS TN 2000, PDP 11-04. WDS: Ca K $\alpha$ , Fe K $\alpha$ , K K $\alpha$ , Na K $\alpha$ . EDS: Mg K $\alpha$ , Al K $\alpha$ , Si K $\alpha$ , Ti K $\alpha$ , Mn K $\alpha$ . 15KV, 30nASC,  $\emptyset \pm 3\mu$ , BC-Integration. Pointanalysis, coating: C  $\sim$  200 Å.

*Electron microscope* A scanning electron microscope Cambridge Stereoscan MK II A, fitted with an energy-dispersive X-ray analyser EDAX 707 was used to produce secondary electron images of the fragments and to obtain a qualitative chemical analysis.

*X-ray diffraction* The samples were analysed with a Siemens powder X-ray diffractometer (Cu K $\alpha$ ). The cell parameter  $a_0$  of coexisting spinels was calculated from the lattice spacings (110); the (220) and (440) lattice spacings were used for confirmation.

### 3. Morphology

The contact between the black coating and the sherd body is sharp (plate 1). The coating is 15–30  $\mu$ m thick, isotropic and contains some small frequently elongated irregular pores. Its surface is rough and shows small rounded particles (plate 2). Some open craters and closed bulbs can be noticed. The whole aspect is characteristic of a coating of the 'glossy clay layer' type and our samples are identical in the macroscopic aspect to those of the Greek black painting (e.g. Noll *et al.* 1973, Noll 1976/77, 1979, Tite *et al.* 1981). By comparison, a glaze has a very smooth surface and would be rich in gas bubbles.

In contrast to the coating, the sherd matrix is very porous and no relict mineral grains can be identified.

### 4. Chemical composition

An initial survey by EDX (energy dispersive X-ray analysis) revealed the presence of Al, Si, K, Fe, with minor Ti (figure 1). In the nomenclature of Noll (1975–1979) the glossy clay layer therefore belongs to the iron reduction or iron-black type of black painting. Quantitative microprobe analysis gave the compositions listed in table 2. The black coating is rich in SiO<sub>2</sub>, Al<sub>2</sub>O<sub>3</sub>, FeO, K<sub>2</sub>O and contains moderate amounts of MgO. In its composition it resembles the classic Attic black coating, of which semiquantitative analyses were published by Brongniart (1844), Hofmann and co-workers (1962–1966) and Noll and co-workers (e.g. 1973). Comparing our results with the complete analysis of Pavicevic (1974) and Tite *et al.* (1981), the higher MgO and K<sub>2</sub>O contents of our sherds become apparent.

The contrast in chemical composition between the gloss and the sherd matrix is pronounced and corresponds to the findings of the authors cited above; the matrix differs markedly in its major elemental chemistry, especially for CaO which is virtually absent in the gloss but high in the sherd.

Some variations between the three coatings analyzed can be noticed for MgO, FeO, CaO and K<sub>2</sub>O. The standard deviations for the oxide contents in a single specimen (very high for MnO and CaO, moderately high for FeO, MgO, Na<sub>2</sub>O and TiO<sub>2</sub>) show the heterogeneous nature of these coatings. This would be expected in the case of a glossy clay layer.

### 5. Mineralogical analysis

The colouring agent of the Attic black pottery has been the object of many studies, and differing pigments have been postulated: *magnetite* and/or *hercynite* (Farnsworth and Wisely 1958,

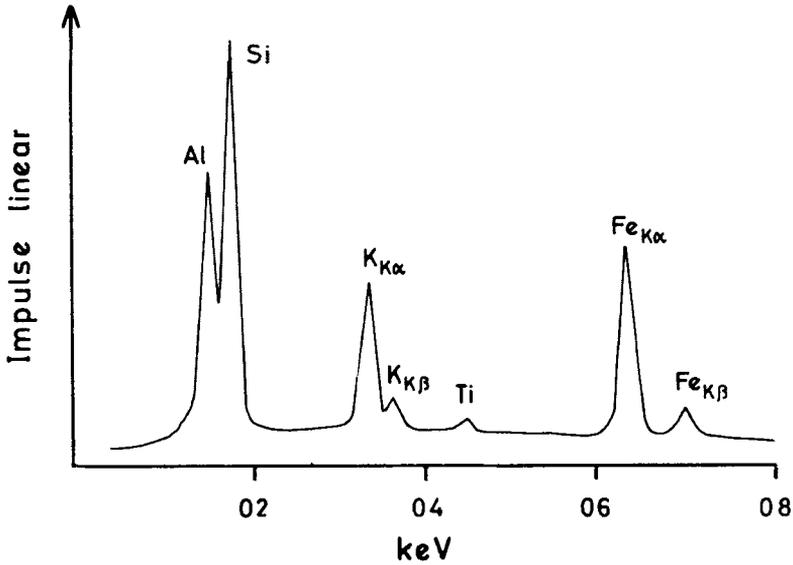


Figure 1 Semi-quantitative analysis of a black coating by EDX.

Table 2 Chemical composition of the black coating (mean and standard deviation) of the Campanian pottery and the Athenian black (Pavicevic 1974). Analysis by electron microprobe.

	KP 1 (n = 7)			KP 5 (n = 3)			KP 10 (n = 9)			Athenian black
	x	s	s%	x	s	s%	x	s	s%	
SiO <sub>2</sub>	45.89	0.64	1.4	45.06	0.30	0.7	45.02	0.27	0.6	43.12
TiO <sub>2</sub>	0.41	0.09	23.6	0.32	0.02	6.2	0.48	0.02	5.2	0.43
Al <sub>2</sub> O <sub>3</sub>	29.12	0.44	1.5	28.86	0.07	0.2	28.80	0.28	0.9	29.00
FeO*	15.64	0.88	5.7	14.09	0.75	5.3	14.40	0.69	4.8	14.15
MnO	0.09	0.07	75.6	0.11	0.02	18.2	0.12	0.07	57.5	0.02
MgO	3.17	0.21	6.7	2.21	0.13	5.9	2.35	0.18	7.6	1.69
CaO	0.17	0.03	17.5	0.65	0.08	12.3	0.76	0.19	25.6	0.35
Na <sub>2</sub> O	0.83	0.05	6.8	0.88	0.08	9.1	0.94	0.08	8.3	n.d.
K <sub>2</sub> O	4.89	0.20	4.1	6.98	0.17	2.4	6.74	0.25	3.7	3.69

\* total Fe as FeO, x = mean value, s = standard deviation, n.d. = not determined.

Hofmann and Russow 1963, Hofmann and Theisen 1965a, 1965b, Hofmann 1966, Hofmann *et al.* 1966, Noble 1960, Noll 1979, Oberlies and Koeppen 1953, 1962), *wuestite* (Bimson 1956), *hypersthene* (Pavicevic 1974).

The surface of 11 objects was studied by X-ray diffractometry. Two phase associations have been identified:

- hercynite + magnetite + hematite + quartz + sanidine (figure 2)
- hercynite + magnetite + quartz + sanidine.

An amorphous phase is to be expected. Cristobalite could not be detected. The relative amounts



Plate 1 *Contact zone between the black coating and the porous sherd matrix. Sherd no. KP 1.6-3259/11.  
Bar = 40  $\mu\text{m}$ .*

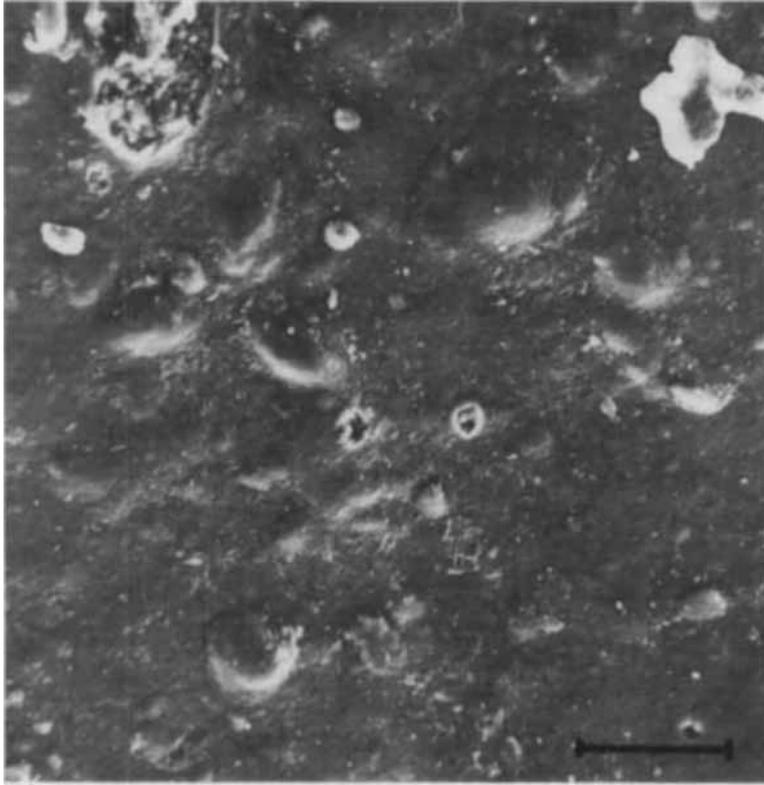


Plate 2 *Surface of a black coating. Specimen no. KP 1.6-3268/2. Bar = 20  $\mu$ m.*

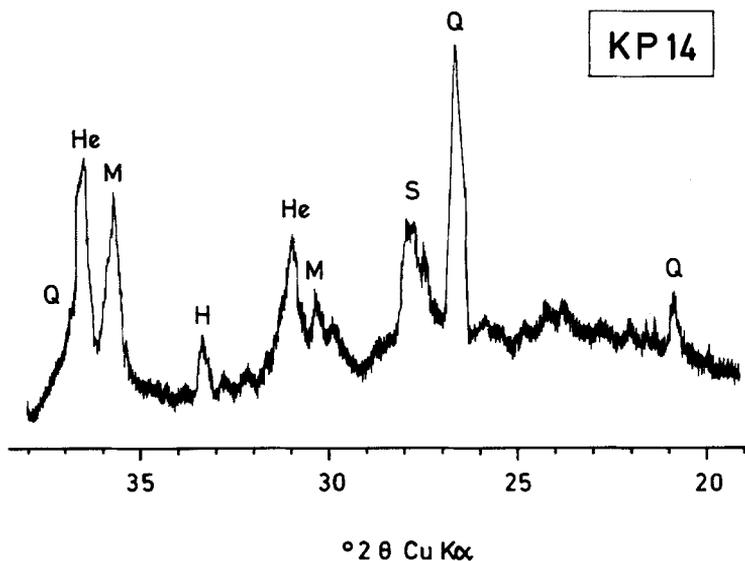


Figure 2 X-ray diffractogram of a black coating (surface X-rayed) composed of hercynite (He), hematite (H), magnetite (M), quartz (Q) and sanidine (S). Sample KP 14.

Table 3 Cell parameter  $a_0$  of coexisting spinel phases in the black coating and chemical composition after Turnock and Eugster (1962, figure 4)

specimen	$a_0$ (Å)		% hercynite in magnetite	% magnetite in hercynite
	hercynite	magnetite		
KP 1	8.172	8.390	—	8
KP 5	8.164	8.364	12	3
KP 6	8.163	8.341	21	3
KP 10	8.169	8.362	14	6
KP 14	8.162	8.328	30	4

of these minerals vary greatly, especially the magnetite/hercynite ratio. The measurements of the cell parameter  $a_0$  for the coexisting spinel phases are given in table 3. They are in close agreement with those of Hofmann *et al.* (1966) and Noll (1979).

A rough approximation of the composition of these coexisting spinels can be made on the basis of the experimental data of Turnock and Eugster (1962). According to these authors a certain amount of MgO, TiO<sub>2</sub> and Fe<sub>2</sub>O<sub>3</sub> must be expected. However, the particles considered here are too small for an exact determination of their chemical composition by microprobe.

## 6. Technology

The firing technique of the Attic potters used for the production of the famous Attic black pottery was elucidated among others by Bimson 1956, Oberlies and Koeppen 1953, Schumann 1942, Winter 1978. The pottery was produced in a single firing during which a first oxidizing

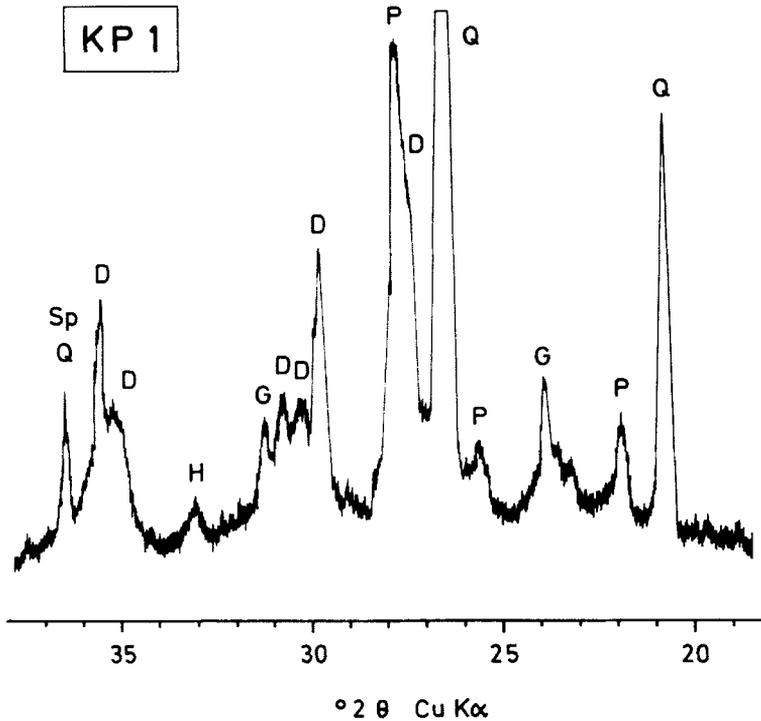


Figure 3 X-ray diffractogram of the powdered sherd matrix from sample KP 1. D = diopside, G = gehlenite, H = hematite, P = plagioclase, Q = quartz, Sp = spinel.

phase up to 800°C was followed by a reducing one (maximum at 900–950°C) with a final cooling process under oxidizing conditions.

The firing temperature of our samples can be estimated by the mineralogy of the sherd bodies, assuming that the phase association present in the sherds reflects the one formed at high temperature and that no important changes occurred during cooling or during burial. The analysed samples KP 1, 5 and 10 have the same phase assemblage:

diopside/wollastonite + gehlenite + quartz + plagioclase + hematite + spinel (figure 3).

The problem of gehlenite in ceramic bodies has been treated by Heimann and Maggetti (1980) and Maggetti and Heimann (1979), who showed that the presence of gehlenite depends primarily upon the grain-size of the initial clay. Illite could be detected in some of the other 17 sherds from the same locality, indicating therefore that a calcareous illitic clay was used for this pottery. Based on the refiring experiments of Kuepfer and Maggetti (1978), Maggetti (1980a, 1980b) and Peters and Jenni (1973) a firing temperature of at least 900°C must be assumed. The absence of illite in some specimens may indicate that the upper limit was slightly higher (950–1000°C). This is in good agreement with the assumptions made for the Athenian black ware (800–950°C: Binns and Frazer 1929, 800–950°C: Farnsworth and Wisely 1958, 825–950/975°C: Noble 1960, 850–1050°C: Tite *et al.* 1981, 830–850°C: Winter 1959/1978).

Some assumptions about the reducing conditions can be made by the coexistence of two Fe-spinels (magnetite<sub>ss</sub> + hercynite<sub>ss</sub>) in the black coatings. Turnock and Eugster (1962)

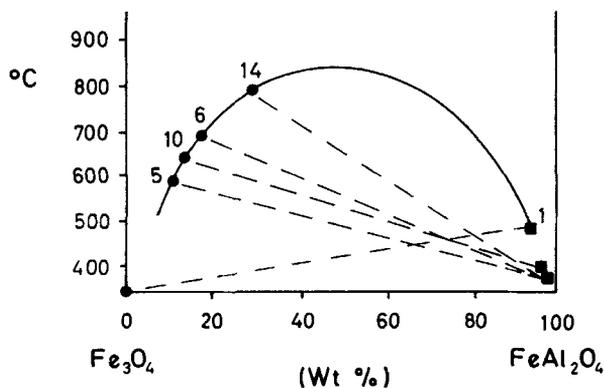


Figure 4 Solvus relations of the X-rayed spinels in the diagram of Turnock and Eugster (1962). Dots = magnetite<sub>ss</sub>(Fe<sub>3</sub>O<sub>4</sub>), squares = hercynite<sub>ss</sub>(FeAl<sub>2</sub>O<sub>4</sub>). Numbers = specimen.

showed that both spinels are separated from the complete solid solution spinel phase stable at higher temperatures by a solvus with a convolute temperature of  $860 \pm 15^\circ\text{C}$  for 48 wt.% FeAl<sub>2</sub>O<sub>4</sub>. The presence of two spinel phases is therefore due to a unmixing process on cooling involving a homogeneous spinel formed at the assumed firing temperatures of  $900\text{--}950^\circ\text{C}$  during the reduction firing phase. By plotting the composition of the coexisting spinels on the solvus of Turnock and Eugster (1962) the equilibration temperature of both phases could in principle be established. However, our specimens give contrasting temperatures, i.e.  $600\text{--}800^\circ\text{C}$  for the magnetite<sub>ss</sub>,  $350\text{--}500^\circ\text{C}$  for the hercynite<sub>ss</sub> (see figure 4) indicating that both spinels are not in equilibrium. This is probably connected with the fact that the exact chemical composition of both spinels could not be determined. Alternatively, the rapid cooling did not allow equilibration.

According to Turnock and Eugster (1962, figure 12a), the absence of wuestite and corundum in the black coating points to oxygen fugacities in the range of  $-12$  to  $-21 \log_{10} f_{\text{O}_2}$  for the magnetite<sub>ss</sub>; the 'equilibration temperature' of the hercynite<sub>ss</sub> is too low for a correct interpolation.

An indication that the reoxidation phase also affected the black layer is given by the presence of hematite in some specimens which certainly formed on the outer parts of the gloss.

The *mineralogical nature* of the unfired glossy clay layer used for the production of the Attic black vases and the Roman terra sigillata remains a matter of debate. Some authors assume that special additions i.e. borax or potassium carbonate were made to a fine grained clay suspension (Neumann 1932, Schumann 1942); more recent authors however tend to assume that a natural illitic clay suspension was used without additions (Bimson 1956, Farnsworth and Simmons 1963, Farnsworth and Wisely 1958, Hofmann 1962–1966, Noble 1960, Noll 1973–1979, Oberlies and Koeppen 1953, Pavicevic 1974, Tite *et al.* 1981, Willgallis and Heyer 1976, Winter 1959, 1978).

The comparison of our analyses with the published illite compositions shows agreement for all elements with the exception of SiO<sub>2</sub> which is lower and FeO which is higher. The high FeO-content is characteristic of the Attic black and the terra sigillata red and it indicates the presence of an iron-oxide or -hydrate in the slip, present naturally or added artificially. The coatings of Rheinzaubern are similar to our samples with respect to the low SiO<sub>2</sub>-content. By comparison with the results of Willgallis and Heyer (1976) as well as Tite *et al.* (1981) it can be

assumed with some certainty that our coatings too were initially a highly ferrous, illitic clay suspension. A test on boron by emission spectroscopy was negative, so that at least the addition of borax is excluded.

Another question is the *origin* of the gloss; was it made locally from the same calcareous clay used for the production of the sherd bodies or did the ancient potters use another, imported low calcareous clay? The contrasting chemical nature between gloss and matrix, especially in CaO content, points to differences in the initial clays; however a local calcareous illitic clay with coarse calcite grains subjected to decantation could also result in a CaO-free slip. If this assumption is correct, the very high FeO-content of the slip, which was either present naturally in the initial calcareous clay or was added in the form of Fe-oxides or -hydrates, must be explained. Without a definite knowledge of the possible clay sources of the Calès region where this pottery was found no attempt can be made to solve this problem.

#### ACKNOWLEDGEMENTS

The authors would like to thank Mrs O. Marbacher for chemical analyses as well as Mrs M. Bruegger for photographic work, Mr A. Jornet for the drawings and Mrs G. Pillier for typing the manuscript. We are grateful to Mr H. U. Nissen who greatly improved the manuscript's English.

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